

## A STUDY OF THE Pt-Al-Ru SYSTEM AT 600°C

S.N. Prins,\* L.A. Cornish\*\* and P.S. Boucher\*

\* CSIR-National Metrology Laboratory, PO Box 395, Pretoria, 0001, South Africa

\*\* Physical Metallurgy Division, Mintek, Private Bag X3015, Randburg, 2125, South Africa

The Pt-Al-Ru system is being studied as part of a larger project to develop and optimise Pt-based alloys for high temperature use [1]. These alloys are based on a two-phase microstructure of  $\sim\text{Pt}_3\text{Al}$  in a (Pt) matrix, analogous to the  $\gamma/\gamma'$  microstructure of Ni-based superalloys. Work has been done on the Pt-Al-Ru system [2,3] and the liquidus surface has been derived from as-cast alloys [3].

Six alloys were selected from the alloys so as to contain the phases of interest. The samples were sealed in silica tubes backfilled with argon and annealed at 600°C for 3 weeks. They were prepared metallographically and studied with a LEO 1525 SEM and Oxford INCA EDS. The phases were confirmed, as far as possible, using a Philips XRD with Cu K alpha radiation on solid samples.

The  $\sim\text{Pt}_{51}:\text{Al}_{21}:\text{Ru}_{28}$  sample comprised coarse needles of (Ru) in a binary eutectic of fine (Ru) needles and  $\sim\text{Pt}_3\text{Al}$ . Compared to the as-cast sample, the fine needles had coarsened, and there were no traces of the (Pt) component. Thus the heat treatment had removed the ternary eutectic which appeared due to non-equilibrium cooling. There was precipitation of  $\sim\text{Pt}_3\text{Al}$  in the coarse (Ru) needles; this indicated that the (Ru) solvus slopes to lower Ru contents at lower temperatures, and agrees with Obrowski's observations in the Al-Ru system [4].

The as-cast  $\sim\text{Pt}_{25}:\text{Al}_{46}:\text{Ru}_{29}$  sample comprised very cored  $\sim\text{RuAl}$  dendrites in a matrix of  $\sim\text{PtAl} + \text{Pt}_5\text{Al}_3$  which had originated from solid state decomposition of the high temperature beta phase. The heat treated sample showed much reduced coring (Fig. 1) and coarsening in the matrix phases.

The  $\sim\text{Pt}_{39}:\text{Al}_{52}:\text{Ru}_9$  sample in the as-cast condition had a complex structure that revealed primary formation of cored  $\sim\text{RuAl}$  followed by the formation of PtAl and  $\text{Pt}_2\text{Al}_3$ . The actual reactions were difficult to interpret since the PtAl and  $\text{Pt}_2\text{Al}_3$  phases were extremely fine. Annealing at 600°C reduced the coring in  $\sim\text{RuAl}$  and coarsened the microstructure so that a eutectic between  $\sim\text{RuAl}$  and  $\sim\text{PtAl}$  was revealed. The  $\sim\text{PtAl}$  within the eutectic had a higher Ru content, and so had a slightly darker contrast as indicated by the arrow in Fig. 2.

As-cast  $\sim\text{Pt}_{14}:\text{Al}_{54}:\text{Ru}_{32}$  was another complex sample and was not at equilibrium since it contained four phases:  $\sim\text{RuAl}$ ,  $\sim\text{RuAl}_2$ ,  $\sim\text{PtAl}_2$ , and  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$ , a new ternary phase [3]. The annealed sample only had three phases:  $\sim\text{RuAl}$ ,  $\sim\text{RuAl}_2$  and  $\sim\text{PtAl}_2$ . In addition, there was precipitation of  $\sim\text{RuAl}_2$  within  $\sim\text{PtAl}_2$

The as-cast  $\sim\text{Pt}_{28}:\text{Al}_{64}:\text{Ru}_8$  specimen contained dendrites of  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$  surrounded by  $\sim\text{PtAl}_2$ , in a eutectic comprising  $\sim\text{PtAl}_2$  and  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$ . In the annealed condition, there was much less of the  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$  phase and the eutectic had coarsened.

In the as-cast condition, the  $\sim\text{Pt}_8:\text{Al}_{85}:\text{Ru}_7$  alloy had two distinct microstructures locally and the primary phase was different in each:  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$  and  $\sim\text{Pt}_5\text{Al}_{21}$  respectively. The other phases were

$\sim\text{RuAl}_6$  and (Al). Although the annealed sample contained regions which appeared different, the  $\sim\text{Pt}_5\text{Al}_{21}$  phase had disappeared, and the  $\sim\text{RuAl}_6$  phase was not discerned. However, since the  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$  phase still showed coring, it is likely that the  $\sim\text{RuAl}_6$  phase was still present and was in local equilibrium with the less Pt-rich composition of  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$ , but too fine to detect.

The phase and alloys' EDS analyses were plotted and compared to the as-cast values. The alloys suffered minimal aluminium loss on annealing.  $\text{Pt}_3\text{Al}$  had lost all discernible Ru, which agrees with other work [2]. Similarly,  $\text{RuAl}_2$  had negligible Pt after annealing, showing that the solubility for Pt decreases with temperature. The composition of  $\sim\text{Ru}_{12}\text{Pt}_{15}\text{Al}_{73}$  moved to slightly lower Pt contents at lower temperatures. Two samples exhibited a similar and a higher Ru composition for the  $\sim\text{PtAl}_2$  phase than in the as-cast samples, indicating that the solubility increased with temperature. Both the  $\text{PtAl}$  and  $\text{Pt}_2\text{Al}_3$  phase compositions shifted to more stoichiometric values after annealing, indicating a contraction in phase width at lower temperatures. At  $600^\circ\text{C}$ , the penetration of the  $\sim\text{RuAl}$  phase was reduced compared to the as-cast samples: from  $\sim 26$  at. % Pt to  $\sim 22$  at. % Pt. In addition, the phase width narrowed at lower temperatures.

Annealing the samples at  $600^\circ\text{C}$  equilibrated them to some degree; no sample had more than three phases, and the compositions had changed to more stoichiometric values. The only unexpected result was that the  $\sim\text{PtAl}_2$  phase extended to higher ruthenium contents.

## References

- [1] L.A. Cornish, J. Hohls, P.J. Hill, S.N. Prins, R. Süß and D.N. Compton, 34<sup>th</sup> International October Conference on Mining and Metallurgy Proceedings, Ed. Z.S. Markovic and D.T. Zivkovic, 545-550, 30 September - 3 October 2002, Bor Lake, Yugoslavia.
- [2] T. Biggs, P.J. Hill, L.A. Cornish and M.J. Witcomb, *J. Phase Equilibria*, 22 (2001) 214-215.
- [3] S.N. Prins, L.A. Cornish, P.S. Boucher and W.E. Stumpf, submitted to *J. Alloys and Compounds*.
- [4] W. Obrowski, *Metallwissenschaft und Technik* (Berlin), 17 (1960) 108-112.
- [5] This research was supported by the PDI and the DS&T, South Africa.

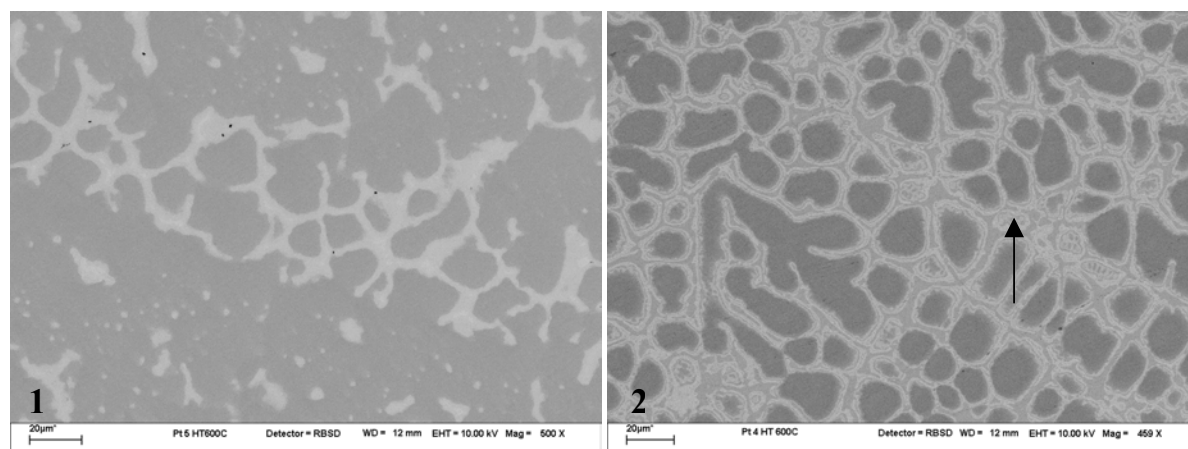


Fig. 1.  $\sim\text{Pt}_{25}:\text{Al}_{46}:\text{Ru}_{29}$ : BSE image showing  $\sim\text{RuAl}$  (dark) in a matrix of  $\sim\text{PtAl} + \text{Pt}_5\text{Al}_3$  (light).  
 Fig. 2.  $\sim\text{Pt}_{39}:\text{Al}_{52}:\text{Ru}_9$ : BSE image showing  $\sim\text{RuAl}$  (dark),  $\sim\text{PtAl}$  (light) and  $\text{Pt}_2\text{Al}_3$  (medium grey).