# EFFECTS OF Fe AND V STATES ON THE FENTON CATALYTIC ACTIVITY OF NATURAL RUTILE

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Abstract—As a common mineral phase on Earth and Martian regolith, natural rutile was reported as a potential candidate for use as a Fenton catalyst in this study. The influences of Fe and V in various chemical states on the generation of reactive oxygen species (ROSs) and the catalytic activity of rutile were examined. A series of rutile samples with various surface and bulk states of Fe and V were obtained initially by hydrogen annealing of natural rutile at ~773-1173 K. X-ray diffraction, electron paramagnetic resonance spectra, and X-ray photoelectron spectroscopy demonstrated that the atomic fractions of Fe(III) and  $V(V)$  decreased sharply with increasing temperature, along with the accumulation of surface  $Fe(II)$  and bulk V(III). All as-prepared materials showed enhanced Fenton degradation efficiency on methylene blue (MB) compared with P25-TiO2, and the treated samples exhibited up to 3.5-fold improvement in efficiency at pH 3 compared to the untreated sample. The improved efficiency was attributed mainly to Fenton catalysis involving Fe(II) and V(III). The dissolved  $Fe^{2+}$  played a crucial role in the homogeneous Fenton reaction, while the bound V(III) favored adsorption primarily and may have facilitated heterogeneous Fenton reaction and the regeneration of  $Fe^{2+}$ . The pH regulated the reaction mechanism among homogeneous (pH = 3) and heterogeneous (pH = 3.7) Fenton catalysis and physical adsorption (pH = 5, 6). The aim of the present study was to improve the understanding of the potential role of natural rutile with advanced oxidation functions in Earth systems and even on Mars, which also provide an inspiration for screening natural rutile and any other similar, Earth-abundant, low-cost minerals for environmental application.

Key Words—Fe and V co-doping, Fenton reaction, Heterogeneous catalysis, Homogeneous catalysis, Hydrogen annealing, Natural rutile.

### INTRODUCTION

Natural minerals enjoy many inherent virtues such as large abundance, great variety, low cost, ready availability, and no secondary pollution, making them suitable for large-scale industrial and commercial applications. Natural minerals have also been found to have some environmentally friendly properties in terms of pollutant purification, including surface adsorption (Liang et al., 2010; Uddin, 2017), porous filtration (Dong et al., 2006; Gülay et al., 2014), ionic exchange and immobilization (Griffin et al., 1977; Tekbaş et al., 2008; Gómez-Hortigüela et al., 2014), chemical catalysis such as photocatalysis (Lu et al., 2007; Xia et al., 2013; Li et al., 2018), and Fenton reactions (Matta et al., 2008; Pereira et al., 2012; Xia et al., 2017), as well as mineral–bacteria interactions (Lu et al., 2012; Shi et al., 2016). Of all of the above properties, Fenton and Fentonlike reactions are very efficient and have almost no selectivity towards oxidization (Pignatello et al., 2006; Liang et al., 2010; Xia et al., 2017), and thus are very

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desirable in terms of dealing with severe contamination situations (Pignatello et al., 2006; Matta et al., 2008; Pereira et al., 2012; Laiju et al., 2014). In practice, homogeneous Fenton reactions are active in terms of removing pollutants in bulk solution and heterogeneous Fenton reactions at the mineral surface-liquid interface. In both cases, the catalytic performance is related to the existing states of transition metals in minerals, such as Fe, V, Ti, Cr, Mn, etc. (Liang et al., 2010; Pereira et al., 2012; Liang et al., 2013).

Natural rutile  $(TiO<sub>2</sub>)$  is one of the most common minerals on the Earth's surface and is the most resistant to weathering, always associated with clays, quartz, and feldspar in soil (Jackson et al., 1948; Allen and Hajek, 1989; Wang et al., 2013; Mehmood et al., 2015). Even certain clay deposits, e.g. kaolinite, illite, and bentonite, contain rutile-type TiO<sub>2</sub> (Dolcater et al., 1970; Sayin et al., 1975). Natural rutile was found to exist in Martian soil regolith as well as in the subsurface at an abundance of 0.5-2 wt.% (Baird et al., 1976; Rieder et al., 1997; Ming et al., 2008). As for all natural minerals, complex geological environments endow natural rutile with inherent defects in the form of substituting and interstitial metal ions, vacancies, etc. Among them, Fe and V are the two most common substituting metals and always coexist in rutile crystals (Suzuki and Pavasupree, 2005). Natural Fe- and V-containing rutile has been found to be a cost-effective and environmentally friendly mineral photocatalyst (Lu et al., 2004., 2007; Chuan et al., 2008), but, to date, its Fenton effect has not been reported. The oxidation states of Fe and V are relevant to the Fenton catalytic activity of Fe- or V-doped minerals, such as different kinds of Fe (oxyhydr)oxides and Vdoped magnetite according to Pereira et al. (2012) and Liang et al. (2010a, 2010b, 2013). Taking into account the complexity of the ore-forming environment, e.g. fluid, temperature, and redox potential, the chemical states of originally doped Fe and V in natural rutile from different deposits cannot be the same. Accordingly, rutile samples from different deposits, or which have experienced different weathering actions, should exhibit different catalytic activities, especially in Fenton or Fenton-like reactions. To screen for the most appropriate natural rutile candidate to apply to degradation of pollutants, and to reveal how the surface states of transition metals impact on the environmental functions of rutile in soils, investigation of the influences of the chemical states of Fe and V on the Fenton catalytic activity of rutile is required.

The objectives of the present study were, therefore, to investigate how the diverse states of surface and bulk Fe and V in natural rutile influence its Fenton reaction efficiency. An opportunity was also provided to screen for natural minerals with potential practical value and cost advantages in environmental remediation represented by rutile and its treated products. Moreover, the results revealed here will also help to propose an alternative mechanism of yielding reactive oxygen species (ROSs) by rutile Fenton catalysis on Mars.

## MATERIALS AND METHODS

## Natural material

The natural rutile sample used in the present study was collected from a metamorphism-alteration deposit in Shanxi Province, China. This natural rutile sample, referred to hereafter as untreated rutile, was ground to an average particle size of  $\leq$ 74  $\mu$ m. The general chemical composition of this rutile was  $(Ti_{0.988}V_{0.010})$  $Fe<sub>0.004</sub>$ <sub>1.002</sub>O<sub>2</sub>, as determined using a JXA-8100 electron microprobe (JEOL Ltd., Tokyo, Japan) (Luo et al., 2012).

### Preparation of hydrogen-annealed rutile samples

The untreated rutile in an alumina porcelain boat was loaded into a Lindberg/Blue M high-temperature tubular furnace (Thermo Fisher Scientific, Waltham, Massachusetts, USA) equipped with a vacuum pump. The process of creating vacuum  $(\leq 3.0 \text{ Pa})$  in the furnace and reloading protective argon (Ar) gas was repeated three times. The sample was heated from room temperature (298 K) to the desired annealing temperature (773 K, 873 K, 973 K, 1073 K, or 1173 K, respectively) with a continuous gas flow  $(10\% \text{ H}_2/\text{Ar})$  of 200 mL/min and temperature ramp rate of 15 K/min. During the annealing treatment process, the pressure of the system was maintained at 1.0 atm. After 3 h of annealing, the sample was cooled to room temperature naturally under the Ar atmosphere. The hydrogenannealed rutile samples are referred to hereafter (with reference to their annealing temperatures) as  $H_2$ -773,  $H_2$ -873,  $H_2$ -973,  $H_2$ -1073, and  $H_2$ -1173. The cooled samples were then transferred quickly (several minutes) to an anaerobic glove box filled with Ar gas to avoid oxidation.

### Characterization techniques

Powder X-ray diffraction (XRD) patterns of rutile samples were recorded using a D/max-rA diffractometer (Rigaku Industrial Corporation, Tokyo, Japan) using CuK $\alpha$  radiation at a scanning speed of 0.5°20/min. The accelerating voltage and applied current were 40 kV and 100 mA, respectively. To detect the valence states and local geometric structures of dopant cations, electron paramagnetic resonance spectra (EPR) of untreated rutile and hydrogen-annealed powdered samples with a uniform mass of 0.02 g were measured at liquid nitrogen temperature (77 K) using a JES-FA200 spectrometer (JEOL Ltd., Tokyo, Japan). The spectra were recorded as first-derivative plots at a microwave frequency of 9.1 GHz (X-Band) and power of 1.0 mW. A signal of Mn(II) was used as an internal standard. The X-ray photoelectron spectroscopy (XPS) measurements were carried out using an AXIS Ultras instrument (Kratos Analytical Ltd, Manchester, UK) with  $AIK\alpha$  source (1486.6 eV) to obtain the surface characteristics of dopant ions in rutile samples. The C 1s peak at 284.8 eV served as an internal reference for the calibration of absolute binding energy and all the data were processed using the CasaXPS software package (version 2.3.16, Casa Software Ltd., Teignmouth, UK).

### MB degradation experiments

As a model simulated pollutant, methylene blue (MB) degradation experiments were launched to test the pollutant-removal efficiency of untreated rutile and treated samples. Reactants were combined in an Erlenmeyer flask with a final total volume of 100 mL of suspension in concentrations of 5 mg/L of MB, 1.0 g/L of powdered sample, and 8.8 mmol/L of hydrogen peroxide  $(H_2O_2)$ . The initial pH values were set at some arbitrary points and adjusted by adding HCl or NaOH. Each flask was wrapped in aluminum foil to avoid any influence of light. The reaction system was stirred continuously for 2 h. At 30 min intervals, a 10 mL suspension (10.05 g) was withdrawn from each flask and centrifuged (9900 r/min for 10 min). The MB concentration in the supernatant was determined using an HP-8453 UV-visible spectrophotometer (Agilent,

Santa Clara, California, USA) at the maximum absorbance wavelength of MB (664 nm). The removal rate of MB was calculated using the following formula:

Removal rate 
$$
(\%) = (1 - C_t / C_o) \times 100\%
$$
 (1)

where  $C_0$  is the initial concentration of the dye solution and  $C_t$  is the concentration of the dye solution at time t.

# Measurement of leached Fe and V ions

The concentrations of  $Fe^{2+}$  and total Fe ions leached in solution were measured using the 1,10-phenanthroline spectrophotometric method (Kremer, 2008; Shimizu et al., 2012). Optical absorption was measured at the wavelength of 510 nm (using the aforementioned spectrophotometer), which was the maximum absorption peak of the  $Fe^{2+}$ - $\sigma$ -phenanthroline complex. The experimental conditions were the same as above except that MB was not added in order to avoid color interference in the chromogenic reaction of  $Fe^{2+}$  with 1,10-phenanthroline. After 30 min of stirring, 10 mL of suspension (10.05 g) was removed and centrifuged. Then, the supernatant was subdivided into three equal parts, each of which was used to measure the concentration of  $Fe^{2+}$ , total Fe ions, and V ions. The total Fe ions were all reduced to  $Fe^{2+}$  in the presence of hydroxylamine hydrochloride (NH2HO·HCl) before adding 1,10-phenanthroline. Inductively Coupled Plasma-Atomic Emission Spectrometry (ICP-AES, Teledyne Leeman Labs, Hudson, New Hampshire, USA) was performed to detect the quantity of total dissolved V ions in the reaction solution.

All of the chemicals used in the present study were of analytical purity grade and obtained from Sinopharm Chemical Reagent Co., Ltd, Shanghai, China, including  $H_2O_2$  (30%, V%), methylene blue  $(C_{16}H_{18}N_3SCl·3H_2O)$ , HCl (36%, V%), sodium hydroxide (NaOH), 1,10 phenanthroline  $(C_{12}H_8N_2·H_2O)$ , and hydroxylamine hydrochloride (NH<sub>2</sub>HO·HCl). P25-TiO<sub>2</sub> was purchased from Degussa AG (Frankfurt, Germany). All solutions were prepared at room temperature with 18.2 M $\Omega$ ·cm Mill-Q ultrapure water.

#### RESULTS AND DISCUSSION

#### Mineral phase changes with the annealing temperature

All of the diffraction peaks in the XRD patterns (Figure 1 and inset) were assigned to the phase of rutile- $TiO<sub>2</sub>$  (JCPDS 89-4920) except for an additional diffraction peak  $(44.6°2\theta)$  for high-temperature-annealed samples  $(H_2-1073$  and  $H_2-1173)$ . This peak, indicated with an arrow, was assigned to the strongest diffraction peak of zero-valence metallic iron  $(Fe<sup>0</sup>)$  for the crystal face (110) (JCPDS 89-7194). The formation of  $Fe<sup>0</sup>$  indicated that Fe in untreated rutile had been reduced to lower valence states (<+3) after hydrogen treatment. Nevertheless, rutile was still the main mineral phase before and after the treatment.

# Variation of Fe and V in surface and bulk states

EPR. The mass fractions of Fe and V in the rutile samples as shown by EMPA results were 0.4 wt.% and 1.0 wt.%, respectively (Luo et al., 2012). For such low dopants, EPR is the most effective technique for determining the valence states, occupied sites, and local symmetries of Fe(III), Fe(II), V(IV), V(III), and V(II) due to the existence of unpaired electrons (Occhiuzzi et al., 2003; Gopal et al., 2004). Each EPR



Figure 1. XRD traces of untreated rutile and hydrogen-annealed samples: (a) untreated rutile, (b)  $H_2$ -873, (c)  $H_2$ -973, (d)  $H_2$ -1073, (e)  $H<sub>2</sub>$ -1173. The arrows indicate the (110) crystal face of iron (enlarged in inset image, upper right). The XRD trace of  $H<sub>2</sub>$ -773 is not shown.



Figure 2. EPR spectra of untreated rutile and hydrogen annealed samples recorded in the magnetic field from 0 to 260 mT. The g values labeled in the figure correspond to EPR signal peaks caused by  $Fe(III)$  located in various symmetry sites in rutile-TiO<sub>2</sub>.

signal of Fe(III) in untreated rutile and hydrogen annealed samples (Figure 2) was assigned to the local structure of Fe(III) based on the differences in g-values (Table 1). For untreated rutile, Fe(III) was the main state and Fe(III) symmetry sites had two types. First, the signal at  $g = 4.23$  was typical of an isolated rhombic Fe(III) cation in a high-spin configuration with a strong distorted environment  $(E/D = 1/3)$  (Thorp and Eggleston, 1985; Soria et al., 1991; Egerton et al., 2001; Pecchi et al., 2003; Chakradhar et al., 2006). Here D and E are the axial and rhombic component parameters of the crystal field, respectively. The second group with peaks at g = 7.99, 5.58-5.02, 3.44, and 2.62 in untreated rutile correspond to signals for substituting Fe(III) with a relatively lower rhombic distortion (Amorelli et al., 1987; Soria et al., 1991; Pecchi et al., 2003).

After hydrogen annealing, only three peaks ( $g = 5.61$ , 4.92, and 2.62) remained for the modified samples  $(H_2$ -773,  $H_2$ -873,  $H_2$ -973, and  $H_2$ -1073), the intensities of which decreased sharply. Compared with natural rutile, the narrower and more symmetrical EPR peaks implied that the substituting Fe(III) should be located at a more stable crystal structure, indicating that high temperature probably enhanced the mineral crystallinity. In addition, the vanishing of EPR signals and decline of intensities with the increasing temperature suggested the decrease of unpaired electrons outside the Fe electronshell, which could be ascribed to Fe(III), was gradually reduced to lower valence states, *i.e.* Fe(II) and Fe<sup>0</sup>. In terms of  $H_2$ -1173, no EPR signal was detected except a curved baseline over the entire range of magnetic field, attributed to the formation of  $Fe(II)$  and  $Fe<sup>0</sup>$ , both of which are silent in conventional EPR measurement (Knapp et al., 2000).

EPR signals of vanadium in untreated rutile and hydrogen-annealed samples (Figure 3) revealed that all the samples except  $H_2$ -1073 and  $H_2$ -1173 exhibited an obvious eight-fold hyperfine splitting structure. The calculated g and A (the hyperfine coupling constants) values  $(g_{//} = 1.98, g_{\perp} = 1.93, A_{//} = 15.4 \text{ (mT)}, \text{ and}$  $A_{\perp}$  = 4.1 (mT)) were very close to that of substituting V(IV) in rutile (Gallay et al., 1986; Kera and Matsukaze, 1986; Cavani et al., 1988; Rodella et al., 2002; Tian et al., 2009). For untreated rutile, judging from the weak EPR signal of  $V(IV)$ ,  $V(V)$  could be the main state of V accompanying the small amount of V(IV), which were demonstrated in detail in previous work (Luo et al.,

Table 1. EPR signal assignments for the untreated rutile and hydrogen-annealed samples in the magnetic range from 0 to 260 mT (Thorp and Eggleston, 1985; Amorelli et al., 1987; Soria et al., 1991; Egerton et al., 2001; Pecchi et al., 2003; Chakradhar et al., 2006).

Sample	Experiment g value	Assignment
Untreated rutile	4.23 7.99, 5.58, 5.02, 3.44, 2.62	Fe(III) located in a strong distorted rhombic symmetry site $(E/D = 1/3)$ $Fe(III)$ substituted for $Ti(IV)$ in the rutile crystal lattice with less rhombic distortion
$H_{2}$ -773, $H_{2}$ -873, $H_2-973$ , $H_2-1073$	5.61, 4.92, 2.62	$Fe(III)$ substituted for $Ti(IV)$ in the rutile crystal lattice at a more stable structure



Figure 3. (a) EPR spectra of untreated rutile,  $H_2$ -773, and  $H_2$ -873; (b) EPR spectra of  $H_2$ -973,  $H_2$ -1073, and  $H_2$ -1173 with  $\times$  4 magnification recorded in the magnetic field from 264 to 500 mT. The \* symbol represents the signal from the internal Mn(II) standard. The  $g_{//}$  and  $g_{\perp}$  marked on the figure correspond to eight-fold hyperfine splitting structure lines of V(IV).

2012). After hydrogen annealing, the intensity of signal assigned to  $V(IV)$  notably strengthened in the  $H<sub>2</sub>$ -773 sample, but then diminished sharply in  $H<sub>2</sub>-873$  and  $H<sub>2</sub>$ -973 with the rising temperature. In the case of the H2-1073 sample, the eight-fold hyperfine splitting structure could not be identified clearly due to the low intensity. For  $H<sub>2</sub>$ -1173, the eight-fold hyperfine splitting structure had disappeared completely, which could result from the formation of V(III) which had no EPR signal, as for Fe(II) (Brückner, 2006). Based on the discussion above, the substantial increase in V(IV) at 773 K was due to the reduction of  $V(V)$  and then  $V(IV)$  gradually converted to V(III) above 773 K.

Under the present experimental conditions (77 K, X-band, microwave frequency (~9.5 GHz)), neither Fe(II) nor V(III) was detected directly because they displayed considerable zero-field splitting (ZFS) due to the absence of unpaired electrons  $(3d^6, 3d^2)$  (Knapp et  $al., 2000; Br\"uckner, 2006).$  Their signals can only be observed by high-frequency, high-magnetic field EPR or dual-mode EPR. Their existence and variation tendency, however, could be inferred through the vanishing signals of Fe(III) and V(IV).

XPS. X-ray photoelectron spectroscopy is a means of detecting the surface chemical states of Fe, V, Ti, and O

in rutile samples, as well as the atomic ratio, at nanodepths (up to 10 nm). The core-level XPS spectra of Fe 2p in untreated rutile and some treated samples (Figure 4) showed that the binding energy values centered at ~709.0 eV and 722.6 eV (blue fitted lines) were assigned to Fe(II), and those at 711.0 eV and 724.6 eV (green fitted lines) were assigned to Fe(III) (Yamashita and Hayes, 2008). In addition, some satellite peaks, caused by  $Fe 3d - O 2p$  hybridization, appeared at  $\sim$ 714.7 eV (blue line) and 718.8 eV (green line) corresponding to Fe(II) and Fe(III), respectively (Fujii et al., 1999; Yamashita and Hayes, 2008). As the temperature increased, the Fe(II) signal strengthened and this could be interpreted as an increase in Fe(II) as a proportion of total Fe. The XPS signals of  $Fe<sup>0</sup>$  should emerge at Fe  $2p_{3/2} = 706.9$  eV and Fe  $2p_{1/2} = 719.9$  eV, but these were undetected in all samples (Fiedor et al., 1998; Geng et al., 2009). This phenomenon might be due to the tiny particle size, small content, and uneven distribution of Fe<sup>0</sup>, making it difficult to detect. Based on the integrated area of Gaussian-resolved peaks and the sensitivity factor for Fe (Wagner, 1983), the atomic fractions of Fe(II) and Fe(III) in the detected volume under different conditions were obtained (Table 2). The ratio of surface Fe(II)/Fe(III) remained relatively stable at  $\sim 0.3$  in the lower annealing temperature range

Table 2. Surface Fe species (at.%) in rutile samples.

Chemical states	Untreated	$H_{2} - 773$	$H_{2} - 873$	$H_{2} - 973$	$H_{2} - 1073$	$H_{2} - 1173$
Fe(II)	0.32	0.48	0.50	0.64	0.58	0.09
Fe(III)	0.95	l.65	1.74	1.24	0.54	0.11
$\Sigma$ (Fe(II) + Fe(III))	. .27	2.13	2.24	1.88	1.12	0.20
Fe(II)/Fe(III)	0.34	0.29	0.29	0.52	1.07	0.82



Figure 4. Fe 2p core-level XPS spectra of some rutile samples. The black lines were experimental data with poor signal-to-noise ratio due to low content. The blue and green lines are fitted peaks of Fe(II) and Fe(III), respectively. The red lines represent simulative curves based on fitted results.

 $(T \le 873 \text{ K})$ , but began to increase (>0.5) at higher temperatures  $(T \geq 973 \text{ K})$ . This considerable change suggested that the surface Fe(III) was reduced to Fe(II). Note that the total atomic fraction of surface Fe(II) and Fe(III) in hydrogen-annealed samples, *i.e.*  $H_2$ -773 (2.13 at.%),  $H_2-873$  (2.24 at.%), and  $H_2-973$ (1.88 at.%), were obviously greater than that of the untreated rutile  $(1.27 \text{ at.} \%)$ , indicating the migration of Fe from the bulk to the surface in high-temperaturetreated samples. As for  $H_2$ -1073 and  $H_2$ -1173, the substantial decrease of total ionic Fe might be due to the formation of  $Fe<sup>0</sup>$ , according to the XRD results. Combining with the EPR results, the conclusion is that both surface and bulk Fe(III) were reduced almost entirely to Fe(II) and Fe $^0$  at 1173 K.

As for surface V states, no observable V-related signal was detected in untreated rutile but weak signals were found in hydrogen-annealed samples (Figure 5). Although the mass fraction of V in the original rutile sample was 1.5 times greater than Fe (Luo et al., 2012), the atom percentage of surface V  $(0.3 \text{ at} \%)$  detected in all treated samples was much lower than that of Fe (1.0 at.%). The V atoms, therefore, preferred to stay in

the bulk rather than at the surface site, even after hightemperature annealing. Under high temperature and the reducing atmosphere of hydrogen, only a few V species could migrate towards the surface. After smoothing and fitting processes (red curves in Figure 5), the peak centers and assignment of binding energy values to V  $2p_{3/2}$  were obtained (Table 3). All treated samples exhibited very weak peaks centered from 516.5 eV to 515.7 eV, which were assigned to the valence of V from V(V) to V(III) (Sawatzky and Post, 1979; Wagner, 1979; Luo et al., 2012). Note that neither a  $V(II)$  nor a  $V(I)$ signal was observed in samples for which the V  $2p_{3/2}$ binding energy values were <515 eV, implying that V(III) was the terminal state of the hydrogen-modified rutile series. The binding energy values for V  $2p_{3/2}$ decreased gradually with elevated temperature, which confirmed that V underwent reduction reactions. Another interesting phenomenon was that the XPS results showed that 1073 K was the temperature at which surface V(III) formed, but the EPR results implied that V(III) might start to appear at 873 K. The difference here was attributed to the fact that XPS was able to obtain just the elementary information at the surface,



Figure 5. V  $2p_{3/2}$  core-level XPS spectra of all rutile samples, under various treatment conditions. The solid black lines represent the experimental spectra and the red solid lines, the smoothed curves. The dotted black line connects the crest of every curve demonstrating the changing trend of oxidation states, *i.e.* from  $V(V)$  to  $V(III)$ .

while EPR demonstrated the valence-state information at the surface and in bulk. This result showed that V(III) produced at 873 K would not be present at the surface of sample until the temperature reached 1073 K, further indicating that V(III) could migrate from bulk to surface gradually from 873 K to 1073 K. The different behaviors of V and Fe might be related to their chemical mobility and diffusivity.

In addition, the surface chemical states of Ti and oxygen vacancy throughout the whole reduction process should lead to other concerns. Surprisingly, Ti(IV) was the only observed Ti state (Figure S1 in Supplementary Material, available from http://www.clays.org/Journal/ JournalDeposits.html) and the atomic fraction of oxygen vacancy didn't change much in any rutile samples (Figure S2 and Table S1 in Supplementary Material).

Another problem was how V(V) was reduced to V(III) when most V did not really exist at the surface sites and could scarcely be reduced by hydrogen directly. In fact, these problems are attributed to the same issue. The reduction of V(V) in the bulk phase could benefit from a small amount of Ti(III) which was first reduced by hydrogen on the rutile surface. Then, charge transfer occurred from surface to bulk through linkages of V-O-Ti, followed by subsequent reduction of  $V(V)$  and regeneration of Ti(IV) (Trifir, 1998; Klosek and Raftery, 2001; Bhattacharyya et al., 2008). Such a process is referred to as inter-valence charge transfer (IVCT) and is described in equation 2. Besides, the ability of Fe(III) to accept electrons was greater than that of Ti(IV) and similar IVCT could help to minimize the reduction of Ti(IV), as is expressed in equation 3 (Seki

Table 3. V  $2p_{3/2}$  binding energy values and corresponding valences for all rutile samples.

	<b>Samples</b>					
	Untreated	$H_{2} - 773$	$H_{2} - 873$	$H_{2} - 973$	$H_{2} - 1073$	$H_{2} - 1173$
Peak center (eV) Assigned valence	$\overline{\phantom{m}}$	516.5 $V(V)$ , $V(IV)$	516.3 V(IV)	516.1 V(IV)	515.9 $V(IV)$ , $V(III)$	515.7 V(III)

-: No V XPS signal was detected in the untreated samples.

et al., 2005). The absence of Ti(III) could further avoid oxygen atoms escaping from the lattice, maintaining them in the Ti(IV) $O_6$  octahedron. In this regard, V(V) and Fe(III) could serve as a coupled redox buffer to keep the stability of the rutile phase even in severely reduced conditions.

$$
Ti(III) + V(V) \rightarrow V(IV) + Ti(IV)
$$
 (2)

$$
Ti(III) + Fe(III) \rightarrow Fe(II) + Ti(IV)
$$
 (3)

According to EPR and XPS results, Fe(III) and V(V) were the predominant impurity cations in untreated rutile. After hydrogen annealing, on one hand, a predictable tendency was observed that Fe(III) migrated gradually from the bulk to the surface and was concurrently reduced to lower oxidation states, i.e. Fe(II) and Fe $^0$ . On the other hand, V(V) was also transformed into V(IV) and V(III) after hydrogen treatment, but they were located mainly in the bulk rather than at the surface. Based on the above discussion, the main thermal chemical reactions during the hydrogen annealing process are summarized as follows in equations 4 and 5:

873–1073 K:  
\n
$$
(Ti_{0.988}(V^V, V^{IV})_{0.01}Fe_{0.004}^{III})_{1.002}O_2 + H_2 \rightarrow
$$
\n
$$
(Ti_{0.988}(V^{IV}, V^{III})_{0.01}Fe_x^{II}Fe_{0.004-x}^{II})_{1.002}O_2
$$
\n(4)

# 1173 K:  $(\text{Ti}_{0.988}(\text{V}^{\text{IV}}, \text{V}^{\text{III}})_{0.01} \text{Fe}^{\text{II}}_{x} \text{Fe}^{\text{III}}_{0.004-x})_{1.002}\text{O}_2 + \text{H}_2 \rightarrow$  $(Ti_{0.988} V_{0.01}^{III} F e_{0.004-y}^{II}) O_2 + y F e^0$  (5)

where x and y represent the ratio of newly formed  $Fe(II)$ and elemental Fe, respectively  $(0 \le x \le 0.004, 0 \le y \le 0.004)$ .

# MB degradation efficiency by hydrogen-annealed samples

The MB was used as a simulated pollutant to check the chemical activity of all prepared rutile samples as well as  $P25-TiO<sub>2</sub>$  and a blank control (no catalyst) (Figure 6). The removal of MB in the blank control was negligible whether  $H_2O_2$  was added or not; in contrast, rutile samples worked well for the removal of MB. In the system with  $H_2O_2$  at pH = 3.0, the untreated rutile sample showed a much greater rate of degradation of MB (27.4%) than did P25-TiO<sub>2</sub> (4.7%). Because no substituting transition metal cation was found in P25-  $TiO<sub>2</sub>$ , the result implied that the substituting Fe and V in the rutile sample played a key role in MB degradation. In addition, compared to untreated rutile, the hydrogenannealed rutile samples displayed a significantly enhanced removal rate of MB as the annealing temperature increased from 773 K to 1173 K (illustrated as black columns in Figure 6). The two highesttemperature annealed samples  $(H_2-1073$  and  $H_2-1173)$ showed a ~2.5 times greater MB-removal rate than that of untreated rutile, suggesting samples with lowervalence metal species (Fe(II) and V(III)) favored promotion of reaction activities. Compared to  $H_2$ -1073, the MB degradation efficiency by  $H_2$ -1173 decreased slightly, which could be ascribed to the reduction of total electrovalent Fe in  $H_2$ -1173 due to the formation of Fe<sup>0</sup>.

The physical adsorption of MB in the absence of H2O2 was achieved in parallel (gray columns in Figure 6). The adsorption rates of all samples after 2 h were <20%. The same tendency was exhibited as for those systems with  $H_2O_2$ , *i.e.* the lower the valence state of the cations, the greater the removal efficiencies they



Figure 6. MB-removal efficiencies under multiple experiment conditions (black and gray columns represent reaction systems with and without  $H_2O_2$ , respectively) for untreated rutile and hydrogen samples annealed at various temperatures, as well as P25-TiO<sub>2</sub> and a blank control (in the absence of catalyst). The total reaction time was 2 h. Composition of reaction mixture (100 mL final volume): sample, 1.0 g/L; initial MB concentration, 5.0 mg/L;  $H_2O_2$  concentration, 8.8 mmol/L; initial pH: 3.

had. In particular,  $P25-TiO<sub>2</sub>$  showed the worst adsorption performance even though it had a large surface area and good adsorption capability. The results not only confirmed the improved degradation capability towards MB of natural rutile and its treated samples in the presence of  $H_2O_2$ , but also suggested that those surface V and Fe cations were indeed considerable factors in the adsorption of MB molecules. In conclusion, the decrease in MB concentration should be due mainly to the chemical reaction which occurred between Fe/V codoped rutile samples and  $H_2O_2$ , as well as physical adsorption.

### Discussion of the roles of Fe(II) and V(III)

According to the experimental results above, the enhanced MB degradation rate could be attributed to the combined effect of  $H_2O_2$  and cation-bearing materials. This complies exactly with the characteristics required for Fenton reaction in which  $H_2O_2$  is converted to a hydroxyl radical (·OH) in the presence of reduced-state cations (Liang et al., 2010; Pereira et al., 2012; Liang et al., 2013). In order to verify how Fe and V got involved in this reaction, it was necessary to measure their corresponding dissolved cations. In the experimental solution after 0.5 h, measurement results (Table 4) showed that the concentration of  $Fe^{2+}$  and  $Fe^{3+}$  in the solution of H<sub>2</sub>-1173 system were  $\sim$ 3 and  $\sim$ 4 times greater than that of untreated rutile, respectively. Clearly, the dissolved Fe<sup>2+</sup> could react with  $H_2O_2$  and further produce  $Fe^{3+}$  and  $·OH$ ; the latter had powerful oxidizing ability to degrade organic pollutants (Wang, 2008). This is a so-called homogeneous Fenton reaction and is the primary reason for the high rate of MB degradation by sample  $H<sub>2</sub>$ -1173, according to equation 6:

$$
\text{Fe}_{\text{aq}}^{2+} + \text{H}_2\text{O}_2 \rightarrow \text{Fe}_{\text{aq}}^{3+} + \cdot \text{OH} + \text{OH}^-,
$$
  
\n
$$
E_0 = 0.528 \text{ V (at pH = 3)}
$$
 (6)

The concentration of dissolved V ions in each suspension after reaction, however, was below the detection limit (ppb level), indicating that hardly any V ion was leached out. As mentioned above, the adsorption effect was enhanced for the treated sample which was due to the fact that the atomic fractions of surface Ti(IV) and oxygen vacancy remained constant, and to the considerable leaching of Fe ions. The stable bonding of V on the surface of rutile, especially V(IV)

and V(III) in H<sub>2</sub>-973, H<sub>2</sub>-1073, and H<sub>2</sub>-1173 should, therefore, play an important role in physical adsorption. The incorporation of V in  $Fe_{3-x}V_xO_4$  magnetite may promote the adsorption of MB (according to Liang et al., 2010), and the same mechanism could indicate that V on the surface of rutile promoted the adsorption. Besides, the redox potentials for  $H_2O_2/H_2O$ , Fe(III)/Fe(II), and V(IV)/V(III) are 1.299 V, 0.771 V, and 0.337 V (vs. NHE in  $pH = 3$ ), respectively (Bossmann et al., 1998; Liang et al., 2010). Thus, thermodynamics favored  $H_2O_2$ first oxidizing surface V(III) rather than Fe(II). This socalled heterogeneous Fenton reaction took place exactly at the interface between the solid surface and the liquid, which is described in equation 7. Meanwhile, V(III) could contribute to the regeneration of the Fenton active species,  $Fe^{2+}$ , driven by the electromotive force (*i.e.*)  $E_0 = 0.434$  V) (equation 8), just as V(III) worked in V-doped magnetite (Liang et al., 2010a, 2010b, 2013).

$$
V_{\text{surf}}^{\text{III}} + H_2O_2 \rightarrow V_{\text{surf}}^{\text{IV}} + {}^{.0}H + {}^{0}H^-,
$$
  

$$
E_0 = 0.962 \text{ V (at pH = 3)}
$$
 (7)

$$
V_{\text{surf}}^{\text{III}} + Fe_{\text{aq}}^{3+} \rightarrow V_{\text{surf}}^{\text{IV}} + Fe_{\text{aq}}^{2+}, E_0 = 0.434 \text{ V (vs. NHE)} \quad (8)
$$

# Impact of pH on the Fenton effects of  $H_2$ -treated samples

The pH is a very important factor in the Fenton reaction. Comparison of experiments with  $H<sub>2</sub>-1173$  and with an untreated rutile sample were conducted at different pH levels, ranging from 3.0 to 6.0 (Figure 7a).  $H_2$ -1173 and  $H_2$ -1073 showed the greatest MB removal rate at  $pH = 3.0$ , which was much greater than that of untreated rutile and other temperaturetreated samples. The remarkable degrading rate of MB when  $H_2$ -1173 was added conformed to the pseudo-first order law with an apparent rate constant  $k = 1.4609 h^{-1}$ (fitted line shown in inset of Figure 7b). As the pH increased to 3.7, however, the MB removal rate of  $H_2$ -1173 was reduced significantly, to 30.3%. This unsatisfactory degrading result could be attributed to nonhomogeneous catalysis due to the rapidly decreased concentration of leached  $Fe^{2+}$  ions, the activity of which was far less than homogeneous catalysis (Chou et al., 2001; Liang et al., 2010). The approximately linear correlations between the MB removal rate and the reaction time indicated that the reactions involved

Table 4. Concentration of dissolved  $Fe^{2+}$  and total Fe ions for untreated rutile and H<sub>2</sub>-1173.

Sample	Total Fe $(mg/L)$	$\text{Fe}^{2+}$ (mg/L)	Calculated $Fe3+$ (mg/L)
$H_{2} - 1173$	$1.51(\pm 0.009)$	$1.30(\pm 0.036)$	0.21
Untreated rutile	$0.36(\pm 0.047)$	$0.32(\pm 0.099)$	0.04
<b>Blank</b>	$0.022(\pm 0.002)$	bd	0.022

Notes: The concentration data were measured by 1,10-phenanthroline after 0.5 h of reaction at pH 3.0, in the absence of MB and H<sub>2</sub>O<sub>2</sub>. Numbers in parentheses are standard deviation uncertainties. 'bd' means below the detection limit of 0.02 mg/L.



Figure 7. (a) Impact of pH on the Fenton reaction of the  $H_2$ -1173 sample and untreated rutile by degrading MB after 2 h. (b) MB removal rate of H<sub>2</sub>-1173 corresponding to various pH values as a function of reaction time. Inset: fitted by pseudo-first order rate law at pH = 3.0, where the correlation coefficient  $R^2 = 0.97$ .

might follow a zero-order kinetics law  $(R^2 = 0.98, \text{ not}$ shown). Physical adsorption, therefore, might play a vital role because, in this situation, the reaction rate depended only on the concentration of the substance adsorbed or adsorption sites but had nothing to do with MB concentration. Whether the pH was set at 5.0 or 6.0, H2-1173 and untreated rutile kept the same trends of enhanced MB removal rate (Figure 7a,b). These performances may be ascribed to the physical adsorption and/or heterogeneous Fenton reaction involving surface bonding of Fe and V, as efficient homogeneous Fenton catalysis was impossible due to the lack of dissolved cations. In these two cases, the increasingly quick reaction rate happened after 1 h, suggesting that two reactions were involved, i.e. a physical reaction followed by a chemical reaction, conforming exactly to the characteristics of heterogeneous catalysis which usually exhibits activity in neutral pH conditions (Liang et al., 2010; Pereira et al., 2012; Liang et al., 2013). The best performance in terms of eliminating a pollutant in strong acidic conditions means that natural rutile and its treated samples have the potential to solve extreme environmental problems. The material used in the present study, i.e. natural rutile and hydrogen-treated rutile, might also suggest other natural minerals to be used in the same way to deal with environmental problems.

### GEOLOGICAL IMPLICATIONS

Natural rutile has been suggested to exist on Earth since the era of planetary accretion (>4.55 Ga) (Hazen et al., 2008). Besides, rutile has been identified in Martian regolith and in the subsurface at an abundance of 0.5-2 wt.% (Baird et al., 1976; Rieder et al., 1997; Ming et al., 2008). On inanimate Mars, the enrichment of reactive oxygen species (ROSs), e.g. hydroxyl radical ( $\cdot$ OH) or superoxide radical ( $\cdot$ O<sub>2</sub>), is assumed to be the main cause of 'lifelessness' because of destructive

effects on organic matter (Lasne et al., 2016). Deep insight into the formation pathways of ROSs and their reactivity on Mars, therefore, is central to understanding prebiotic biochemistry and this will also be helpful in the exploration and preservation of biomarkers. The remarkable role of  $TiO<sub>2</sub>$  in the photocatalytic formation and accumulation of ROSs when commercially synthetic  $TiO<sub>2</sub>$  was used as a reactant was described by Zent et al. (2008). Natural rutile must be imperfect with nonstoichiometry and impurities (such as substitutional Fe and V) in the actual geological circumstance in question, e.g. in Martian soil; this will distinguish it from synthetic, purchased  $TiO<sub>2</sub>$  which has fewer lattice defects. More importantly, the valence of V in Martian melt and minerals was suggested as a mixture of both  $V(III)$  and  $V(IV)$ , while Fe(II) and Fe(III) were both likely species of Fe which vary with oxygen fugacity (Papike et al., 2005; Karner et al., 2006). As a result, to construct more analogous Martian surroundings in the laboratory on Earth, natural rutiles with low valence states of cations, like  $H_2$ -973,  $H_2$ -1073, and  $H_2$ -1173, rather than a commercial rutile, are probably better raw materials. On the other hand, a trace of  $H_2O_2$  was observed directly in the Martian atmosphere and with a global average mixing ratio ranging from 18 to 32 ppb; this can penetrate up to hundreds of meters into the regolith (Bullock et al., 1994; Zent, 1998; Clancy et al., 2004; Encrenaz et al., 2004). Consequently, the interaction between natural rutile and  $H_2O_2$  in the form of the Fenton effect, as is presented in this work, should be a considerable pathway to produce ROSs; this, however, was not mentioned or was ignored in previous studies of Mars. In the present work, rutile samples with V(III), V(IV), Fe(II), and Fe(III) as foreign impurities (namely  $H_2$ -973,  $H_2$ -1073, and  $H_2$ -1173) exhibited a quite striking ability to eliminate MB molecules compared with commercial  $P25-TiO<sub>2</sub>$ , which provided another potential and efficient pathway to generate strong

oxidizing ·OH and impact on the prebiotic biogeochemistry process.

### **CONCLUSIONS**

In the present study, the influences of the chemical states of Fe and V on the Fenton reaction activity of a natural rutile sample were investigated. By hydrogen annealing from 773 K to 1173 K, the surface  $Fe(II)$ increased due to hydrogen reduction and the migration of bulk Fe to the surface by the heating effect. At the same time, V experienced reduction from  $V(V)$  to  $V(III)$ but, compared to Fe, favored remaining in the bulk lattice. As a result, the modified rutile samples displayed substantially increased MB removal rate due to the homogeneous Fenton effect of dissolved  $Fe<sup>2+</sup>$ , physical adsorption, and heterogeneous Fenton effect of V(III). In addition, the efficiency and mechanism of MB degradation depended heavily on pH. Strong acidic conditions were beneficial to homogeneous Fenton catalysis, while faintly acid and neutral conditions favored physical adsorption and a heterogeneous Fenton reaction, respectively. The V and Fe co-doped rutile, in particular with low valence state cations, may promote the Fenton catalytic activity. The present study verified the Fenton effect of natural Fe- and V-bearing rutile and identified the crucial roles of V and Fe with different valence states. This could stimulate inspiration to utilize natural minerals and provide strategies for modifying minerals in the field of environmental science. The present research presents a novel pathway to ·OH generation, which might have happened or be happening on barren and lifeless Mars.

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