THE STABILITY OF Fe-Mg CHLORITES IN HYDROTHERMAL SOLUTIONS: II. THERMODYNAMIC PROPERTIES

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Abstract—The hydrothermal stabilities of a low-Fe clinochlore and a high-Mg chamosite, in the presence of kaolinite, were investigated recently at $T \leq 200^{\circ}$ C and $P_v = P_{H_2O}$ (Aja and Small, 1999; Aja and Dyar, 2002). Standard state thermodynamic properties $(S_{298}^0, \Delta_f H_{1,298}^0 \text{ and } \Delta_f G_{1,298}^0)$ have been obtained for the two chlorites whose structural formulae are $(Al_{2,33}Fe_{1,0}^2Fe_{0,14}^4Ca_{0,02}Mn_{0,01}Ni_{0,02}Cr_{0,01}Mg_{8,40}\square_{0,07})$ (Si_{5.66}Al_{2.34})O₂₀(OH)₁₆ and $(Fe_{0,66}^{+}Fe_{5,43}^{+}Mg_{2,30}Al_{2,98}Mn_{0,05}Ca_{0,03}Zn_{0,01}\square_{0,60})$ (Si_{5.63}Al_{2.37})O₂₀(OH)₁₆. For the low-Fe clinochlore, the respective thermochemical properties are 430 J mol⁻¹ K⁻¹, -8770.64±35.24 kJ mol⁻¹, and -8120.54 ± 32.63 kJ mol⁻¹. $\Delta_f H_{1,298}^0, \Delta_f G_{1,298}^0$ and S_{298}^0 , similarly obtained for the Windsor chamosite are -7851.29 ± 23.14 kJ mol⁻¹, -7271.01 ± 21.43 kJ mol⁻¹ and 668 ± 5 J mol⁻¹K⁻¹, respectively. Ideal site-mixing models of chlorite composition, along the chamosite-clinochlore binary, fail to model satisfactorily these chlorite-fluid equilibria only at lower temperatures ($T < 175^{\circ}$ C). The magnitudes of the excess thermodynamic properties calculated for these chlorites, within the ternary clinochlore-daphnite-sudoite system, suggest significant deviations from ideality.

Key Words—Chamosite, Clinochlore, Solid-solution, Sudoite, Thermodynamic Properties.

INTRODUCTION

A number of activity-composition models have been employed to model compositional variations in chlorites (Stoessell, 1984; Walshe, 1986; Holland et al., 1998; Vidal et al., 2001). Stoessell formulated a random, regular-solution site-mixing model based on a set of six end-member component phases (amesite [(Mg₄Al₂) $(Si_2Al_2)O_{10}(OH)_8]$, chamosite $[(Fe_4^{2+}Al_2)(Si_2Al_2)O_{10}]$ $(OH)_{8}$], Fe³⁺-chamosite [(Fe₄²⁺Fe₂³⁺) (Si₂Al₂)O₁₀(OH)₈], talc-3 brucite [(Mg₆)(Si₄)O₁₀ (OH)₈]), minnesotaite-3 $Fe(OH)_2[(Fe^{2+}_{6})(Si_4)O_{10} (OH)_8]$, pyrophyllite-2 gibbsite $[(Al_4)(Si_4)O_{10}(OH)_8])$; these end-members were ideal 14 Å chlorites for which thermochemical properties were not known. Activity-compositional models for chlorites have also been framed in terms of ideal sitemixing models (Helgeson and Aagaard, 1985); this presumes that compositional variation in the solidsolution is not accompanied by exchange of atoms between energetically distinct sites, and that the distribution of atoms is random on energetically equivalent sites. The ideal site-mixing model has been used rather extensively (Walshe, 1986; Jahren and Aagaard, 1992; Saccocia and Seyfried, 1994). Walshe (1986), for instance, implemented an ideal site-mixing model for chlorites from hydrothermal systems based on a set of six end-member components (Mg₆Si₄O₁₀(OH)₈, $(Mg_5Al_2Si_3O_{10}(OH)_8, (Fe_5^{2+}Al_2 Si_3O_{10}(OH)_8, (Fe_5^{2+}Fe_2^{3+})_{10}(OH)_{10})_{10}(OH)_{10}$ $Si_{3}O_{10}(OH)_{8}$, $(Al_{4}Si_{4}O_{10}(OH)_{8}$, $(Fe_{4}^{2+}Fe_{4}^{3+}Al_{2}Si_{3}O_{11})$ $(OH)_7$). By contrast, Holland *et al.* (1998) proposed a model which permits mixing of atoms on energetically distinct structural sites and a variable degree of cation

* E-mail address of corresponding author: emenike_nduzo@msn.com ordering amongst the end-members: Al-free chlorite $(Mg_6Si_4O_{10}(OH)_8, clinochlore [Mg_6Si_4O_{10}(OH)_8], ame$ site [(Mg₄Al₂)(Si₂Al₂)O₁₀(OH)₈] and daphnite $[(Fe_5^{2+}Al)(Si_3Al)O_{10}(OH)_8]$. This model, proposed in an effort to harmonize predicted and experimental compositional dependence on temperature for the chlorite + forsterite + orthopyroxene assemblage under elevated P-T conditions, presumed a rather simple chlorite chemistry $[(R_{6-x}^{2+}Al_x)(Si_{4-x}Al_x)O_{10}(OH)_8]$ and the absence of octahedral vacancies in the 2:1 layer. But as noted by Vidal et al. (2001), octahedral vacancies in chlorites formed at temperatures <450°C are not artifacts of mixed layering. Hence, they proposed a 3-site mixing model having symmetric Margules parameters and ideal inter-site interaction. Their model is applicable to a wider P-T range and presumes that octahedral vacancies result from di-trioctahedral substitutions, that all Fe is divalent, and that compositional variation in natural chlorites may be projected to the compositional space bounded by clinochlore, daphnite, amesite and Mgsudoite $[(Si_3Al)(Al_3Mg_2)O_{10}(OH)_8]$. However, the Margules parameters for interaction on all sites except for M1 were fixed a priori and thus restrict the mixing of the phase components on the M1 site. As is to be expected, some overlap exists in the choice of endmembers for the various activity-composition models though there is much less latitude in the choice of appropriate Fe end-members, especially Fe³⁺ minerals. Moreover, the simplifying assumptions required to implement the more robust models (Holland and Powell, 1996; Holland et al., 1998; Vidal et al., 2001) attest to the persistent need for additional experimental determinations of thermochemical properties of natural chlorites.

In a prior contribution, Aja and Dyar (2002) presented the results of the experimental investigations of the relative stability of natural Fe-Mg chlorites (a low-Fe clinochlore and a high-Mg chamosite) in hydrothermal solutions. These experimental studies demonstrated that: (1) chlorite-fluid equilibrium is attainable under low temperature and/or diagenetic conditions; (2) appropriate solubility models for chlorite-fluid equilibria may be developed using the law of mass action (solubility constant approach); and (3) slopes of chemical potential diagrams may constrain the identity of solubility-limiting phases. These compositionally complex natural chlorites behaved as singlephase, single-component micas of fixed compositions during solution equilibration experiments (see Appendix). But because chlorites are compositionally complex minerals whose behavior over the course of geological time, as opposed to the time-frame of an experimental investigation, may be best viewed as a solid-solution, it is instructive to examine the implications of these recent low-temperature solution equilibration data for chlorite solid-solution. In this contribution, therefore, standard state thermodynamic properties will be retrieved for the two natural chlorites and the implications of the solution equilibration data will also be evaluated in terms of the ideal site mixing model of chlorite solid solution.

RESULTS

The overall structural formulae for the two chlorites used in the study are $(Al_{2,33}Fe_{1.00}^{2+}Fe_{0.14}^{3+}Ca_{0.02}Mn_{0.01}$ $Ni_{0.02}Cr_{0.01}Mg_{8.40}\Box_{0.07})(Si_{5.66}Al_{2.34})O_{20}(OH)_{16}$ and $(Fe_{0.60}^{3+}Fe_{5.43}^{2+}Mg_{2.30}Al_{2.98}Mn_{0.05}Ca_{0.03}Zn_{0.01}\Box_{0.60})$ $(Si_{5.63}Al_{2.37})O_{20}(OH)_{16}$. These compositions conform to the general formulae, $(\Box_z R_u^2 R_y^{3+})(Si_{4-x}Al_x)O_{10}(OH)_8$

where the number of octahedral vacancies (z) is given by (y - x)/2 and u + y + z = 6 (Wiewióra and Weiss, 1990). Thus, whereas the low-Fe clinochlore is clearly a tritrioctahedral chlorite (z < 0.5; Wiewióra and Weiss, 1990), the identity of the Windsor chamosite (di-tri-, or tri-dioctahedral chlorite) is not resolved owing to lack of detailed knowledge of the distribution of the octahedral vacancy. A putative structural chemistry for the chlorites (Table 1) presumes, in the first instance, that the M4 and T2 sites were fully occupied by Al and Si and that all octahedral vacancies are assignable to the M1 site (cf. Vidal *et al.*, 2001). The atomic proportion of Fe^{2+} in the M1 site was then estimated using Mössbauer spectroscopic data (Aja and Dyar, 2002) and the balance of the M1 site occupancy was assigned to Mg. The M1 site assignments invariably pre-dispose the occupancies of the M2+M3 site inasmuch as M4 occupancy is somewhat certain. Though vacancies may well exist in the interlayer hydroxide sheet (Wiewióra and Weiss, 1990) and Mössbauer data cannot be relied upon to ascertain Fe site assignments (Aja and Dyar, 2002), the assignments nonetheless reflect the prevailing understanding of chlorite chemistry (cf. Welch et al., 1995; Holland et al., 1998).

Equilibrium constants retrieved from the solubility data, as a function of temperature, have been summarized in the Appendix (Table A1); details of experimental techniques and results have been reported previously (Aja and Small, 1999; Aja and Dyar, 2002). The apparent free energy of formation of the chlorites under isothermal, isobaric conditions ($\Delta_a G^0_{chte,T}$) were calculated from the relation,

$$\Delta_a G^0_{chte,T} = -\Delta G^0_{reaction} + \sum_i \Delta_a G^0_{i,T} \tag{1}$$

where $\Delta G_{reaction,T}^0$ and $\sum_i \Delta_a G_{i,T}^0$ are the free energy of reaction (equal to -2.303 RT log K) and the apparent

Putative structural che	mistry c	of two natural cl	nlorites				
	T1	<i>T</i> 2	<i>M</i> 1	<i>M</i> 4	(M2 + M3)		
Low-Fe clinochlore	Si_2	$(Si_{0.83}Al_{1.17})$	$(\Box_{0.03}Mg_{0.941}Fe_{0.029}^{2+})$	Al	$(Mg_{3,259}Al_{0.17}Fe_{0.471}^{2+}Fe_{0.07}^{3+}Ca_{0.01}Ni_{0.01}Cr_{0.01})$		
Windsor chamosite	Si_2	(Si _{0.72} Al _{1.28})	$(\Box_{0.30} Fe_{0.342}^{2+} Mg_{0.38}5)$	Al	$(Mg_{0.792}Al_{0.49}Fe_{0.30}^{3+}Fe_{2.378}^{2+}Mn_{0.03}Ca_{0.01})$		
Mole fraction (X) Low-Fe clinochlore					Windsor chamosite		
X _{clinochlore}	0.820				0.200		
$X_{daphnite}$	0.118				0.602		
X _{sudoite}	0.062			0.198			
Gibbs free energy of t	formatio	n from the elem	tients (kJ mol ^{-1})				
	$\Delta_f G_{T,}^0$	$_P = (a - e)T -$	$aT\ln T + d - \frac{bT^2}{2} - \frac{c}{2T};$		$25 \leqslant T \leqslant 200^{\circ}$ C and $P_{v} = P_{H_{2}O}$		
	Low-Fe clinochlore				Windsor chamosite		
$a \times 10^2$		-0	.5006		1.7053		
$b \times 10^{-2}$	5.6378			-26.456			
С	-4869312			9053346			
d	18838			-77314			
$e \times 10^2$	2.8980				-9.4564		

Table 1. Some properties of the two natural chlorites.

Gibbs energy of reactants and/or species (Table A2) in the solubility models, respectively. The standard state Gibbs free energy of formation from the elements $(\Delta_f G^0_{chte,T})$ of the reacted chlorites were then calculated using the Gibbs energy function (Robie and Hemingway, 1995) and the relation,

$$\Delta_{f} G^{0}_{chte,T} \supset \Delta_{a} G^{0}_{chte,T} - \left[\sum_{j>1}^{n} \left(\frac{G^{0}_{T} - H^{0}_{T_{r}}}{T} \right) \bullet T - \sum_{j>1}^{n} \left(\frac{G^{0}_{T_{r}} - H^{0}_{T_{r}}}{T_{r}} \right) \bullet T_{r} \right]$$
(2)

Fit parameters for polynomials defining the functional dependence of $\Delta_f G^0_{T,P}$ on temperatures (at saturated vapor pressures) are given in Table 1. The sequential retrieval of $\Delta_f G^0_{chte,T}$ via $\Delta_a G^0_{chte,T}$ (equations 1 and 2) assures consistency with the geochemical computer code (EQ3/6; Wolery, 1993) used in aqueous speciation analyses (Aja and Dyar, 2002); the latter relies on the revised HKF equations of state for aqueous ions and related thermochemical data (Shock and Helgeson, 1988; Johnson *et al.*, 1992).

DISCUSSION

Generalized solubility model

That a multiplicity of solubility models is needed to rationalize the behavior of the two chlorites in aqueous solutions is evident from Table A1 (Appendix) and reflects the deliberate omission of oxygen and pH buffers in the experimental charges as well as the effects of temperature on the properties of aqueous electrolytes, lability of reactions, aqueous geochemistry of iron and trace element content of the chlorites. Certainly, the multiplicity of the models yields important insights into the complexity of the reaction mechanisms but it also precludes a strict comparison of the thermal properties of the reactions. The latter requires derivation of a generalized solubility model for each chlorite (equation 3) from the experimental data; *i.e.*

$$\begin{split} (Al_{2.33}Fe_{1.00}^{2+}Fe_{0.14}^{3+}Ca_{0.02}Mn_{0.01}Ni_{0.02}Cr_{0.01}Mg_{8.40}\square_{0.07}) \\ &\quad (Si_{5.66}Al_{2.34})O_{20}(OH)_{16} + 19.21H^{+} = \\ 2.335Al_{2}Si_{2}O_{5}(OH)_{4} + 8.40Mg^{2+} + 0.02Ni^{2+} + \\ 1.14Fe^{2+} + 0.02Ca^{2+} + 0.01Mn^{2+} + 0.01Cr^{3+} + \\ &\quad 0.99SiO_{2(aq)} + 12.935H_{2}O + 0.035O_{2(g)} \quad (3a) \end{split}$$

$$\begin{split} (Fe_{0.66}^{3+}Fe_{5.43}^{2+}Mg_{2.30}Al_{2.98}Mn_{0.05}Ca_{0.03}Zn_{0.01}\square_{0.60}) \\ & (Si_{5.63}Al_{2.37})O_{20}(OH)_{16} + 16.83 \ H^{+} = \\ 2.675Al_{2}Si_{2}O_{5}(OH)_{4} + 6.03Fe^{2+} + 2.30Mg^{2+} + \\ & 0.05Mn^{2+} + 0.03Ca^{2+} + 0.01Zn^{2+} + \\ & 0.28SiO_{2(aq)} + 0.15O_{2(aq)} + 11.065H_{2}O \ (3b) \end{split}$$

Equation 3 presumes that Al is conserved in the solid phases and that all redox species exist in the reduced state; though these are standard assumptions, they lead to stoichiometric, and hence mass balance, constraints rather different from those inferred directly from the experimental data (Table A1; Appendix). Inasmuch as thermochemical properties (*e.g.* $\Delta_f G^0_{chte,T}$; Table 1) of the chlorites have been derived from the solubility data, equation 3 thus provides a simplified but realistic basis for evaluating chlorite-kaolinite equilibria under diagenetic conditions. As is evident, the temperature dependence of the equilibrium constants (Figure 1) shows the inherent differences in the thermal characteristics of the reactions of the two chlorites with kaolinite. A visual inspection of the data (Figure 1) suggests a linear dependence of log K₃ on inverse temperature and this is indeed the case for the Windsor chamosite-kaolinite reaction;

$$\log K_{3b} = -49.72 + 20137/T \quad (r^2 = 0.98) \tag{4a}$$

Equation 4a is of the general form,

$$\log K = -\frac{\Delta H_r^0}{2.303 RT} \quad \frac{\Delta S_r^0}{2.303 R}$$
(4b)

For the Windsor chamosite-kaolinite reaction (equation 3b), ΔH_r^0 and ΔS_r^0 are respectively determined as -385.59 kJ mol⁻¹ and -952.14 J mol⁻¹ K⁻¹, respectively. Hence, the 'second law method' yields for the Windsor chamosite, $\Delta_f H_{298,1}^0$ and S_{298}^0 (from $\Delta_f S^0 + \sum_{i=1}^n S_i^0$) equal to -7851.29 kJ mol⁻¹ and 668 ± 5 J mol⁻¹ K⁻¹, respectively; the resulting standard state free energy of formation of the Windsor chamosite (-7267.87 kJ mol⁻¹) is thus consistent with $\Delta_f G_{1,298}^0$ (-7271.01 ± 21.43 kJ mol⁻¹) determined independently of equation 2. By contrast to the Windsor chamosite, the presumption of an inverse, linear temperature-dependence of equilibrium constants for the low-Fe clinochlore-kaolinite reaction (equation 3a, Figure 1) yielded an unrealistic S_{298}^0 . Equation 5 provides a more satisfactory model; *i.e.*

$$\log K_{3a} = 520.87 - 14198/T - 168.20\log T (r^2 = 0.99)$$
(5)

The standard enthalpy of formation for the clinochlore was calculated with the ΔH_r^0 obtained using



Figure 1. Equilibrium constants calculated for generalized chlorite-kaolinite solubility model (equation 3). (a) and (b) represent kaolinite-chlorite equilibrium with the low-Fe clino-chlore and Windsor chamosite, respectively.

equation 5. This, in conjunction with S_{298}^0 obtained by polyhedral summation techniques (430 J mol⁻¹ K), were then used to adjust the standard free energy of formation ($\Delta_f G_{1,298}^0$) obtained using equation 2. The standard state thermodynamic properties ($\Delta_f H_{1,298}^0, \Delta_f G_{1,298}^0, S_{298}^0$) thus derived for the low-Fe clinochlore are -8770.64±32.63 kJ mol⁻¹, -8120.54±21.43 kJ mol⁻¹ and 430 J mol⁻¹ K⁻¹.

The data for the Windsor chamosite (Figure 3b) may certainly be modeled by presuming a non-linear temperature dependence of log K; this, however, presupposes a more complex model (*i.e.* $\Delta C_{p,r} \neq 0$) than is required to explain the data. On balance, a curvilinear dependence of log K3 on inverse temperature over the temperature range studied was expected for both chlorites given that these are rather complex heterogeneous reactions (cf. Nordstrom and Munoz, 1994; p. 222). Apparently, the fact that equation 3 is a net hydrolytic dissolution reaction, rather than a simple dissolution reaction, implies some compensating effects in the thermal properties of the chloritekaolinite reactions and hence the apparent flatness of the function, $\frac{\partial \log K}{\partial (1/T)}$ over a temperature range of up to 200°C.

Ideal site-mixing model of chlorite solid-solution

Mineral compositional variations are amenable to various solid-solution treatments. Of the numerous solid-solution models, the ideal site-mixing model (Helgeson and Aagaard, 1985) has been widely utilized in studies of chlorites and illites found in diagenetic and hydro-thermal assemblages and experimental systems (Bird and Norton, 1981; Walshe, 1986; Aagaard and Helgeson, 1983; Aagaard and Jahren, 1992; Saccocia and Seyfried, 1994). In view of its demonstrated applicability to chlorite-fluid equilibria data at higher temperatures (Saccocia and Seyfried, 1994), it is instructive to evaluate these recent chlorite solution equilibration data with the ideal site-mixing model. The activity (a_i) of the *i*th thermodynamic component of a solid-solution is given by,

$$a_i = \mathbf{k}_i \prod \prod (X_{j,s})^{\mathbf{v}_{s,j,i}} \tag{6}$$

where $X_{j,s}$ is the mole fraction of the *j*th atom on the *s*th crystallographic site in the solid-solution, $v_{s,j,i}$ is the stoichiometric number of the *s*th sites occupied by the *j*th atom in one mole of the *i*th thermodynamic component and k_i is the proportionality constant,

$$\mathbf{k}_{i} = \prod_{s} \prod_{j} (X_{j,s,i})^{\mathbf{v}_{sj,i}}$$
(7)

 $X_{j,s,i}$ is the mole fraction of the jth atom on the sth crystallographic site in the ith stoichiometric mineral. Based on equation 7, the proportionality constant for the low-Fe clinochlore and the Windsor chamosite are 148.60 and 234.34, respectively. Thus, for the thermodynamic components (Mg₅Al)(Si₃Al)O₁₀(OH)₈ and (Fe²₅⁺Al)(Si₃Al)O₁₀(OH)₈,

$$a_i = 148.60(X_{\text{Mg,O}})^5(X_{\text{Al,O}})(X_{\text{Al,T}})(X_{\text{Si,T}})^3$$
 (8a)

$$a_i = 234.34(X_{\text{Fe},O})^5(X_{\text{Al},O})(X_{\text{Al},T})(X_{\text{Si},T})^3$$
 (8b)

For the low-Fe clinochlore, log $a_{\text{clinochlore}}$ and log a_{daphnite} are calculated (equation 8a) to be -0.30 and -4.93, respectively. By contrast, log $a_{\text{clinochlore}}$ and log a_{daphnite} for the Windsor chamosite calculated from equation 8b are -2.81 and -0.94, respectively. Equation 8 implies that the different octahedral sites in the chlorite structure are homologous.

The binary clinochlore-daphnite system provides a framework for modeling chlorite solid-solution (Curtis, 1985; Hutcheon, 1990; Aagaard and Jahren, 1992) and hence the experimentally-measured chlorite-kaolinite reactions (Table A1, Appendix) resolve into the kaolinite-clinochlore and kaolinite-chamosite compatibilities. Hence, at $T \leq 125^{\circ}$ C the appropriate reactions are

$$(Mg_{10}Al_2)(Si_6Al_2)O_{20}(OH)_{16} + 18H^+ = Al_2Si_2O_5(OH)_4 + 2Al(OH)_4^- + 4SiO_{2(aq)} + 10Mg^{2+} + 11H_2O$$
(9a)

and at $T \ge 175^{\circ}$ C,

Though analogous reactions to equation 9 but containing daphnite may be derived easily, only the clinochlore components of the two natural chlorites were evaluated in this work owing to constraints imposed by the experimental data. Specifically, aqueous Fe was rarely present as Fe^{2+} in most of the runs (*cf.* Table A1) owing to the absence of redox potential buffers in the experimental charges and thus precludes a direct treatment of kaolinite-chamosite equilibria.

In Figure 2, values of $log(a_{Mg^2}^{1/2})/(a_H)$ calculated from equation 9, for the chlorite-kaolinite boundary (with the activity of the clinochlore phase components of the natural chlorites, equation 8), have been compared with values measured directly. For experiments conducted with low-Fe clinochlore (Figure 2a), the experimentally-measured kaolinite-chlorite boundary is more stable than the predicted one by nearly six orders of magnitude at 25°C whereas at 200°C, the values differ by ~0.4 log units. A similar trend is observed for the chamosite data (Figure 2b) although at 175° (the highest temperature for which experimental data are available) the values differ by 1.5 log units. In other words, there is a virtual convergence of the experimental and theoretical values at 200°C. This suggests that at $T \ge 200^{\circ}$ C, ideal site-mixing algorithms reasonably model these chloritefluid equilibria in the binary clinochlore-chamosite system. That the ideal site-mixing model satisfactorily predicts clinochlore-kaolinite equilibria under elevated temperature conditions is consistent with the findings of Saccocia and Seyfried (1994) who similarly modeled their higher temperature (300-500°C) chlorite solubility data with an ideal site-mixing model. They concluded



Figure 2. A comparison of predicted and measured $\log(a_{Mg^2}^{1/2})/(a_H)$ for chlorite-kaolinite-fluid assemblage projected to quartz saturation, between 25 and 200°C and at saturated vapor conditions. The predicted values presume an ideal site-mixing model for the chlorites (equations 8–9); smoothed values of the predicted (p) and measured (m) ion activity ratios, as a function of temperature, are indicated by the curves. The standard state thermodynamic properties of Robie and Hemingway (1995) for end-member clinochlore were used in the predictive model; the heat capacity data given in Table 2 were used for extrapolation to higher temperatures.

that an ideal site mixing model may rationalize their solubility data although a non-ideal model with a modest excess Gibbs energy ($\Delta G^{\text{ex}} \approx 837 \text{ J mol}^{-1}$) was not precluded. At lower temperatures, however, the ideal site-mixing model fails to correctly reproduce the experimental data (Figure 2). The failure of the model under standard state conditions (25°C, 1 bar) is particularly significant inasmuch as other natural chlorites (e.g. the Vermont chlorite-kaolinite-quartz assemblage; Kittrick, 1982) equilibrate at a pH $-\frac{1}{2}pMg^{2+}$ value (6.5 at 25°C) comparable to the low-Fe clinochlorekaolinite boundary (Figure 4a; Aja and Small, 1999). In other words, the limited data available indicate that other natural chlorites buffer values of $\log(a_{M\sigma^2}^{1/2})/(a_{\rm H})$ to values comparable to the ones measured in this study. The divergence of the predicted ion activity ratios from experimental data at lower temperatures (Figure 2) could stem from the following factors: metastability of



Figure 3. Variation of calculated excess Gibbs free energies $(kJ \text{ mol}^{-1})$ as a function of temperature. (a) and (b) show $G_{\text{soln}}^{\text{exoln}}$ calculated for the low-Fe clinochlore and Windsor chamosite, respectively.

chlorite-kaolinite assemblages, erroneous standard state Gibbs free energy of formation for end-member chlorites, very large excess Gibbs energy for chlorite solidsolutions under low-temperature conditions or limitations of the ideal site-mixing model. A presumption of metastability of chlorite-kaolinite assemblages under lower temperature conditions is questionable given that the experimental charge should have buffered $\log(a_{\rm Mg^2}^{1/2})/(a_{\rm H})$ to higher, rather than lower, values compared with the predicted values; moreover, the notion that the criteria used to assess equilibrium during the experiments could simultaneously indicate stable (at high temperatures) and metastable (at low temperatures) equilibrium is specious.

The apparent failure of the ideal site-mixing model under low-temperature conditions may also be partially attributed to uncertainties in standard state thermodynamic properties of end-member chlorites (clinochlore or daphnite). Though $\Delta_f G^0_{298,1 \text{ bar}}$ reported by different investigators (cf. Table 2) fall within a standard deviation of 20 kJ mol⁻¹ (1 σ), thermochemical values based on calorimetric data (e.g. Robie and Hemingway, 1995) are somewhat discrepant from those regressed from high-temperature, high-pressure phase equilibrium experiments (Berman, 1988; Holland and Powell, 1990, 1998). For the latter, linear regression approaches yield standard state properties different from those extracted using linear programming (Berman, 1988) techniques. In the former, the refined parameter is $\Delta_t H_T^0$ whereas in the latter all parameters ($\Delta_f H_{t, S_0}^0, V_0$) are minimized. Despite these differences, a presumption of an uncertainty of 20 kJ mol⁻¹ (in $\Delta_f G_{298,1 \text{ bar}}^0$) shifts the calculated ion activity ratios by ~0.65 log units and even at the 2σ levels, a significant difference still exists between measured and predicted ion activity ratios. Indeed, to bring the predicted and measured $\log(a_{\mathrm{Mg}^2}^{1/2})/(a_{\mathrm{H}})$ values into coincidence will require decreasing the

Formula Name	(Mg ₅ Al)(Si ₃ Al)O ₁₀ (OH) ₈ clinochlore	$(Fe_5^{2+}AI)(Si_3AI)O_{10}(OH)_8$ Fe^{2+} -chamosite/daphnite	$(\Box_1 Mg_2 Al_3)(AlSi_3)O_{10}(OH)_8$ sudoite
$\Delta_f H_{1,298}^\circ$	-8919.9 ± 20.0^{a} -8929.86 ± 1.85^{c} -8919.25 ± 4.38^{d}	$-7101.97\pm4.17^{\circ}$ $-7153.99\pm3.33^{\circ}$ $-7148.44\pm4.50^{\circ}$	$-8634.36 \pm 1.86^{\circ}$
$\Delta_f G^0_{1,298}$	-8255.8 ± 20.0^{a} -8263.35^{c} -8250.546^{e}	-6495.13±4.17 ^ь -6535.56 [°]	-7976.77°
$S_{1,298}^{0}$	421.00 ^a 410.50 ^c	559.00 ^b 545.00 ^c	399.00°
$V_{1,298}^0$	21.090	21.340	20.300
Heat capacity	coefficients ($C_{\rm p} = a \times 10^2 + bT - \frac{c}{T}$	$\left(\frac{2}{2}\right)$	
a	6.1279	6.0006	6.5638
$\Delta_{f}a$	0.3918	0.8228	1.0658
b	0.3151	0.3676	0.3843
$\Delta_t b$	0.1396	0.0755	0.2171
c	16873700	14441117	15840039
$\Delta_{f}c$	15652129	15253669	13627740

Table 2. Thermodynamic properties of end-member chlorites*.

* Sources of data: (a) Robie and Hemingway (1995); (b) Saccocia and Seyfried (1993); (c) Holland and Powell (1998); (d) Holland and Powell (1990); (e) Berman (1988). Units: $\Delta_f H_{1,298}^0, \Delta_f G_{1,298}^0, S_{1,298}^0, V_{1,298}, C_p$ are measured in kJ mol⁻¹l, kJ mol⁻¹ J mol⁻¹ K⁻¹, J bar⁻¹, J mol⁻¹ K⁻¹, respectively. Heat capacity taken from Holland and Powell (1998). $\Delta_f a, \Delta_f b, \Delta_f c$ are coefficients for the formation of the end-member chlorites from their constituent elements.

standard state Gibbs free energy of formation of clinochlore by up to 140 kJ mol⁻¹ ($\Delta_f G_{298,1 \text{ bar}}^0 \approx -8396 \text{ kJ mol}^{-1}$).

In addition to uncertainties in standard state thermochemical properties, the inherent limitations of the ideal site-mixing model must be considered in any attempt to rationalize the divergence between predicted and measured ion activity ratios (Figure 2). A fundamental presumption of the ideal site-mixing model is mixing of atoms on energetically-equivalent sites, with consequent resolution of structural sites in chlorite solid-solution into tetrahedral and octahedral sites. This is clearly an approximation considering the multiplicity of cation sites in chlorites (Bailey, 1988; Welch et al., 1995). These cation sites include two tetrahedral sites (T1 and T2), two octahedral sites in the TOT layer [M1](trans) and M2 (cis)] and two octahedral sites in the interlayer (M3 and M4); these four distinct octahedral sites may accommodate Fe²⁺ or Fe³⁺ and atomic site occupancies for the M2 and M3 sites are not easily resolved for chlorites. Moreover, the distributions of Al and Si in tetrahedral sheets of chlorites are apparently highly ordered (Welch et al., 1995). In spite of these structural chemistry imperatives, ideal mixing of atoms on homological sites describes compositional variation in chlorites at high temperatures; but at temperatures characteristic of diagenesis and low-temperature hydrothermal settings, the ideal site-mixing model fails to reproduce the experimental data. This pattern attests to the well established fact that with increased temperatures, there is a greater degree of freedom in the mixing of the components of a solid-solution; i.e. at low temperatures, the contribution of the enthalpy of mixing (ΔH_{mix}) to the Gibbs energy of mixing greatly exceeds the entropy of mixing term $(T\Delta S_{mix})$ suggesting either that a miscibility gap exists in the system, or that the chamosite (daphnite)-clinochlore binary join does not provide an adequate system for modeling diagenetic and/or hydrothermal chlorites. There is thus a need for more robust models, compared to the ideal site-mixing model, of chlorite solid-solutions. Though recent treatments of chlorite solid-solution contain the needed improvements (e.g. Holland et al., 1998; Holland and Powell, 1996; Vidal et al., 2001), an exact implementation of these latter models is constrained by available experimental data. For instance, the recognition that octahedral vacancies are real structural features of low-temperature chlorites (Vidal et al., 2001; Wiewióra and Weiss, 1990) implies that consideration of sudoite (such as Mg-sudoite, $(\Box_1(Mg_2Al_3)(AlSi_3)O_{10}(OH)_8))$ as an end-member mineral is particularly apposite in chlorite solidsolution models developed for low-temperature conditions. If possible contributions of Fe³⁺-chamosite $((Fe_4^{3+}Fe_2^{2+})(Si_2Al_2)O_{10}(OH)_8)$ are justifiably omitted owing to the relatively small amount of Fe³⁺ present in the samples, the activity of the end-member phase components, clinochlore (cln), daphnite (daph) and Mg-sudoite (sud) may be given by (cf. Vidal et al., 2001).

$$a_{\rm daph} = 4(X_{\rm Si}^{T2})(X_{\rm Al}^{T2})(X_{\rm Fe}^{M1})(X_{\rm Fe}^{M2+M3})^4$$
(10a)

$$a_{\rm cln} = 4(X_{\rm Si}^{T2})(X_{\rm Al}^{T2})(X_{\rm Mg}^{M1})(X_{\rm Mg}^{M2+M3})^4$$
(10b)

$$a_{\rm sud} = 64(X_{\rm Si}^{T2})(X_{\rm Al}^{T2})(X^{M1})(X_{\rm Mg}^{M2+M3})^2(X_{\rm Al}^{M2+M3})^2$$
 (10c)

The non-ideal part of the excess function, given by equation 11, becomes

$$RT \ln \gamma_{cln} = (X_{Fe}^2 + X_{Fe} X_{Al}) W_{cln, daph} + (X_{Al}^2 + X_{Fe} X_{Al}) W_{cln, sud} - X_{Fe} X_{Al} W_{daph, sud}$$
(11a)

$$\begin{aligned} \mathsf{R}T \ln \gamma_{\mathrm{daph}} &= (X_{\mathrm{Mg}}^2 + X_{\mathrm{Mg,Al}}) W_{\mathrm{cln, daph}} + \\ & (X_{\mathrm{Al}}^2 + X_{\mathrm{Mg}} X_{\mathrm{Al}}) W_{\mathrm{daph, sud}} - X_{\mathrm{Mg}} X_{\mathrm{Al}} W_{\mathrm{cln, sud}} \end{aligned} \tag{11b}$$

$$RT \ln \gamma_{sud} = (X_{Fe}^2 + X_{Fe} X_{Al}) W_{daph, sud} + (X_{Mg}^2 + X_{Fe} X_{Mg}) W_{cln, sud} - X_{Fe} X_{Mg} W_{cln, daph}$$
(11c)

In equations 10 and 11, X, γ and W represent the relevant mole fractions, activity coefficients and Margules parameters, respectively. A strict application of the above ternary model to the experimental data requires knowledge of the detailed structural chemistry of the chlorite samples particularly the amounts of cation ordering, the extent of intrasite and inter-site mixing within the various sites; these compositional parameters are unknown for the natural chlorites used in this study. Furthermore, the limited experimental data available under isothermal, isobaric conditions (from this investigation) constrain calculation of Margules' parameters even for a symmetric regular solution model. Though some Margules' parameters may be fixed independent of the system under investigation, this circumvents the need to demonstrate the efficacy of the model from first principles. Moreover, the data from this study cannot be augmented with independent solution equilibration data for other natural chlorites (at 200°C \ge T \le 25°C; P_v = $P_{\rm H_2O}$) inasmuch as such complementary data are lacking. Clearly, the need for additional multilateral experimental investigations of chlorite-fluid equilibria under these low-temperature conditions cannot be overemphasized.

Excess thermodynamic functions

That chlorite solid-solutions exhibit significant deviation from ideality under low temperatures is indicated from the foregoing and thus an evaluation of the magnitudes of excess thermodynamic properties (such as Gibbs energy) serves important heuristic purposes. The excess Gibbs energy (G_{soln}^{ex}) for chlorite solid-solution may be obtained from equation 12,

$$G_{\text{soln}}^{\text{ex}} = G_{s.s}^0 - \left(\sum_i X_i G_i^0 - RT \sum_i X_i \ln X_i\right)$$
(12)

where, G_i^0 , $G_{s,s}^0$ and X_i represent the molar Gibbs free energy of the end-members, the molar free energy of formation of chlorite solid-solutions (calculated from experimental data) and mole fraction of the components in the solid-solutions, respectively. Mole fraction (X_i) of the end-member phase components may be defined either as a thermodynamic mole fraction or molecular mole fraction. The former applies if mixing of atoms is random on energetically-equivalent sites whereas molecular mole fraction applies if local charge balance is maintained in the structure of the mineral. As previously noted, cation occupancies in the chlorite structure exhibit short-range ordering and thus validate adoption of molecular mole fraction. $\Delta_f G_{T,P}^0$ for the end-member minerals (clinochlore, daphnite, sudoite) used in this study were calculated from,

$$\begin{aligned} \Delta_{f}G^{0}_{T,P} &= \Delta_{f}G^{0}_{\mathrm{Tr,Pr}} - \Delta_{f}S^{0}_{\mathrm{Tr,Pr}}(T - T_{r}) + \\ \Delta_{f}a[T - T_{r} - T\ln\left(\frac{T}{T_{r}}\right)] + 0.5\Delta_{f}b(2TT_{r} - T^{2} - T^{2}_{r}) + \\ &0.5\Delta_{f}c(T^{2} + T^{2}_{r} - 2TT_{r})T^{-1}T^{-2}_{r} + V^{0}_{1,298}(P - 1) \end{aligned}$$
(13)

In equation 13, Δ_{ta} , Δ_{tb} , Δ_{tc} (Table 2) are the Maier-Kelley heat capacity parameters for the formation of the end-member chlorites from their elements and the other quantities have their usual meanings. The excess Gibbs energy thus calculated (*i.e.* equation 12) for the low-Fe clinochlore and high-Mg chamosite are summarized in Figure 3. In either case, the excess Gibbs energy shows a varying degree of temperature dependence and given that the compositions of each chlorite are constant over the temperature range investigated, these excess functions are apparently dominated by enthalpy contributions. The magnitudes of the excess Gibbs energy imply significant deviation from ideality; the low-Fe clinochlore shows (for the most part) a positive departure from ideality whereas the high-Mg chamosite is characterized by a strongly negative deviation. Though these excess energies apply to molecular solid-solutions rather than strictly to pairwise interatomic interactions, their magnitudes nonetheless suggest rather large Margules parameters.

CONCLUSIONS

In recent experimental investigations (Aja and Small, 1999; Aja and Dyar, 2002), the hydrothermal stabilities of a low-Fe clinochlore and a high-Mg chamosite, in the presence of kaolinite, were investigated under diagenetic conditions ($T \leq 200^{\circ}$ C, $P_v = P_{H_2O}$). Standard state thermodynamic properties (S_{298}^0 , $\Delta_f H_{1,298}^0$ and $\Delta_f G_{1,298}^0$), for these natural chlorites, have been obtained from these solution equilibration data. Equilibrium constants measured for the reactions of these two chlorites with kaolinite have inherently different temperature dependencies; whereas the chamosite-kaolinite reaction shows a linear dependence on inverse temperature, the clinochlore-kaolinite reaction is typified by a curvilinear dependence. In either case, nonetheless, the differential changes in log K concomitant with differential changes in inverse temperature is a rather flat function and results from the compensating effects in the thermal properties of the reactions. At higher temperatures ($T \ge$ 200°C), the experimental data are modeled satisfactorily with an ideal site-mixing model of chlorite solidsolution whereas the same model fails at lower temperatures. Apparently, ΔH_{mix} greatly exceeds the $T\Delta S_{mix}$ term as the temperature is decreased. Application of a more robust, internally consistent solid-solution model is, however, constrained by available data; additional studies designed to advance experimental calibration of chlorite solid-solution models must necessarily include detailed structural chemistry of the natural chlorites being investigated, the use of sulfide fugacity buffers to control redox potential in addition to several components of the present chlorite-fluid equilibration experiments.

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APPENDIX

 $T = 25^{\circ}$ C, 1 bar $\log K_{a1} = 56.18 \pm 6.07$ $(Al_{2.33}Fe_{1.00}^{2+}Fe_{0.14}^{3+}Ca_{0.02}Mn_{0.01}Ni_{0.02}Cr_{0.01}Mg_{8.40}\square_{0.07})(Si_{5.66}Al_{2.34})O_{20}(OH)_{16} + 14.18H^{+} + 0.27O_{2(g)} = 0.016M_{10}O_{$ $Al_{2}Si_{2}O_{5}(OH)_{4} + 8.40Mg^{2+} + 3.66SiO_{2(aq)} + 2.67Al(OH)^{-}_{4} + 0.02Ca^{2+} + 0.01MnO_{4}^{-} + 0.01M$ $0.01 \text{CrO}_{4(aq)}^{2-} + 1.14 \text{Fe}(\text{OH})_{3(aq)}^{0} + 0.02 \text{Ni}^{2+} + 6.04 \text{H}_2\text{O}$ (A1) $T = 75^{\circ}$ C, 1 bar $\log K_{a2} = -23.63 \pm 2.92$ $(Fe_{0.60}^{3+}Fe_{5.43}^{2+}Mg_{2.30}Al_{2.98}Mn_{0.05}Ca_{0.03}Zn_{0.01}\Box_{0.60})(Si_{5.63}Al_{2.37})O_{20}(OH)_{16} + 1.42O_{2(g)} + 4.585H_2O + 1.28H^+ = 0.465H_2O + 0.465H_2$ $Al_2Si_2O_5(OH)_4 + 3.35Al(OH)_4^- + 3.63SiO_{2(aq)} + 2.30Mg^{2+} + 4.52FeOOH +$ $1.51 \text{Fe}(\text{OH})_{3(aq)}^{0} + 0.05 \text{MnO}_{4}^{-} + 0.01 \text{Zn}^{2+} + 0.03 \text{Ca}^{2+}$ (A2) $T = 125^{\circ}$ C, 2.3 bar $\log K_{a3} = 52.11 \pm 6.39$ $Al_2Si_2O_5(OH)_4 + 8.40Mg^{2+} + 1.14Fe^{2+} + 3.66SiO_{2(aq)} + 0.02Ca^{2+} +$ $2.67 \text{Al}(\text{OH})_{4}^{-} + 0.01 \text{Mn}^{2+} + 0.02 \text{Ni}^{2+} + 0.01 \text{CrO}_{4(aq)}^{2-} + 0.028 \text{O}_{2(q)} + 8.905 \text{H}_{2}\text{O}$ (A3) $\log K_{a4} = -15.96 \pm 5.37$ $(Fe_{0.60}^{3+}Fe_{5.43}^{2+}Mg_{2.30}Al_{2.98}Mn_{0.05}Ca_{0.03}Zn_{0.01}\Box_{0.60})(Si_{5.63}Al_{2.37})O_{20}(OH)_{16} + 1.14O_{2(g)} + 3.52H^{+} + 1.395H_{2}O = 0.52H^{-1} + 0.52H^{-1}$ $Al_2Si_2O_5(OH)_4 + 3.35Al(OH)_4^- + 3.63SiO_{2(aq)} + 2.30Mg^{2+} + 4.91FeOOH + 0.011FeOOH + 0.0111FeOOH + 0.011FeOOH + 0.011FFEOOH +$ $1.12Fe^{2+} + 0.05 MnO_4^- + 0.01Zn^{2+} + 0.03Ca^{2+}$ (A4) $T = 175^{\circ}$ C, 8.9 bar $\log K_{a5} = 85.88 \pm 8.44$ $(Al_{2.33}Fe_{1.00}^{2+}Fe_{0.14}^{3+}Ca_{0.02}Mn_{0.01}Ni_{0.02}Cr_{0.01}Mg_{8.40}\square_{0.07})(Si_{5.66}Al_{2.34})O_{20}(OH)_{16} + 16.85H^{+} + 0.27O_{2(g)} = 0.07Mg_{1.01}Ni_{1.02}Cr_{0.01}Mg_{1.01}Ni_{1.02}Cr_{0.01}Mg_{1.01}Ni_{1.02}Cr_{0.01}Ng_{1.01}Ni_{1.02}Ni_{1.02}Ni_{1.02}N$ $Al_2Si_2O_5(OH)_4 + 0.57Fe_2O_{3(s)} + 8.40Mg^{2+} + 0.02Ca^{2+} + 0.02Ni^{2+} + 0.0$ $2.67 \text{AlOOH} + 3.66 \text{SiO}_{2(aq)} + 0.01 \text{MnO}_{4}^{-} + 0.01 \text{Cr}_{4(aq)}^{2-} + 13.09 \text{H}_{2}\text{O}$ (A5) $\log K_{a6} = -13.16 \pm 1.14$ $Al_2Si_2O_5(OH)_4 + AlOOH + 2.35Al(OH)_4^- + 3.63SiO_{2(aq)} + 2.30Mg^{2+} + 2.30Mg^{2+}$ $3.015 \text{Fe}_2 \text{O}_3 + 0.05 \text{MnO}_4^- + 0.03 \text{Ca}^{2+} + 0.01 \text{Zn}^{2+} + 1.94 \text{H}_2 \text{O}$ (A6) $T = 200^{\circ}$ C, 15.5 bar $\log K_{a7} = 62.42 \pm 3.20$ $Al_2Si_2O_5(OH)_4 + AlOOH(s) + 0.01MnOOH(s) + 0.57Fe_2O_3(s) + 1.67Al(OH)_4^- + 0.01MnOOH(s) +$ $3.66 \text{SiO}_{2(aq)} + 8.40 \text{Mg}^{2+} + 0.01 \text{CrO}_{4(aq)}^{2-} + 0.02 \text{Ni}^{2+} + 0.02 \text{Ca}^{2+} + 9.75 \text{H}_2\text{O}$ (A7)

* Solution equilibration data of the two natural chlorites reported elsewhere (Aja and Small, 1999; Aja and Dyar, 2002). For presumptions of solubility models, see Aja and Dyar (2002). Equilibrium constants at a given temperature and pressure (K_T) calculated using the relation, $K_T = \frac{1}{n} \sum_{i=1}^{n} \prod a_i^{v_i}$ where a_i , v_i , n denote the activity of aqueous ions as calculated by EQ3/6, stoichiometric reaction coefficients and the number of data points that constrain the kaolinite-chlorite boundary, respectively.

$\Delta_a G_{TP}^0 = (a - e)T - aT \ln T + d - \frac{bT^2}{2} - \frac{c}{2T}$							
	$a \times 10^2$	$b \times 10^{-2}$	c	d	$e \times 10^2$		
Boehmite	0.8416	-1.1863	-904104	-928846	-4.3373		
Diaspore	0.3334	-3.8972	505722	-922091	-1.4591		
Gibbsite	0.6697	15.0041	639520	-1163248	-3.6142		
Goethite	0.6731	0.3032	-81290	-492818	-3.2244		
Hematite	0.9531	8.3116	1350267	-755920	-4.8781		
Kaolinite	2.6037	16.4886	5924468	-3842580	-13.6558		
$Al(OH)_4^-$	14.3004	-256.38	49924177	-1752468	-75.5312		
Ca ²⁺	4.9391	-110.42	16954830	-724736	-26.3693		
Fe ²⁺	5.8875	-126.54	20723872	-311981	-31.9999		
$Fe(OH)^{0}_{3,(aq)}$	5.3367	-57.87	11708932	-805784	-28.5085		
H ₂ O	0.9317	-3.04	918076	-245807	-4.5690		
CrO_4^{2-}	22.6534	-475.99	97623705	-1503941	-11.9863		
Mn_4^-	9.1470	-177.81	35695632	-703510	-4.6905		
Mg ²⁺	6.5010	-134.26	22893775	-706237	-35.7106		
Ni ²⁺	-19.4496	357.59	-90414425	650148	1042856		
SiO _{2(aq)}	0.5934	1.2149	1679678	-844092	-3.0676		
Zn^{2+}	5.7510	-123.54	19735078	-362823	-31.2937		

Table A2. Apparent Gibbs free energy of formation $(\Delta_a G^0_{T,P,i})$.

 $\overline{\Delta_a G_T^0}$ values (J mol⁻¹) are based on SUPCTR92 (Johnson *et al.*, 1992) and apply to saturation vapor pressures and $T < 250^{\circ}$ C.